

4-30-2021

CO₂-wettability reversal of cap-rock by alumina nanofluid: Implications for CO₂ geo-storage

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[10.1016/j.fuproc.2021.106722](https://doi.org/10.1016/j.fuproc.2021.106722)

This is an author's accepted manuscript of: Ali, M., Aftab, A., Awan, F. U. R., Akhondzadeh, H., Keshavarz, A., Saeedi, A., ... Sarmadivaleh, M. (2021). CO₂-wettability reversal of cap-rock by alumina nanofluid: Implications for CO₂ geo-storage. *Fuel Processing Technology*, 214, article 106722. <https://doi.org/10.1016/j.fuproc.2021.106722>

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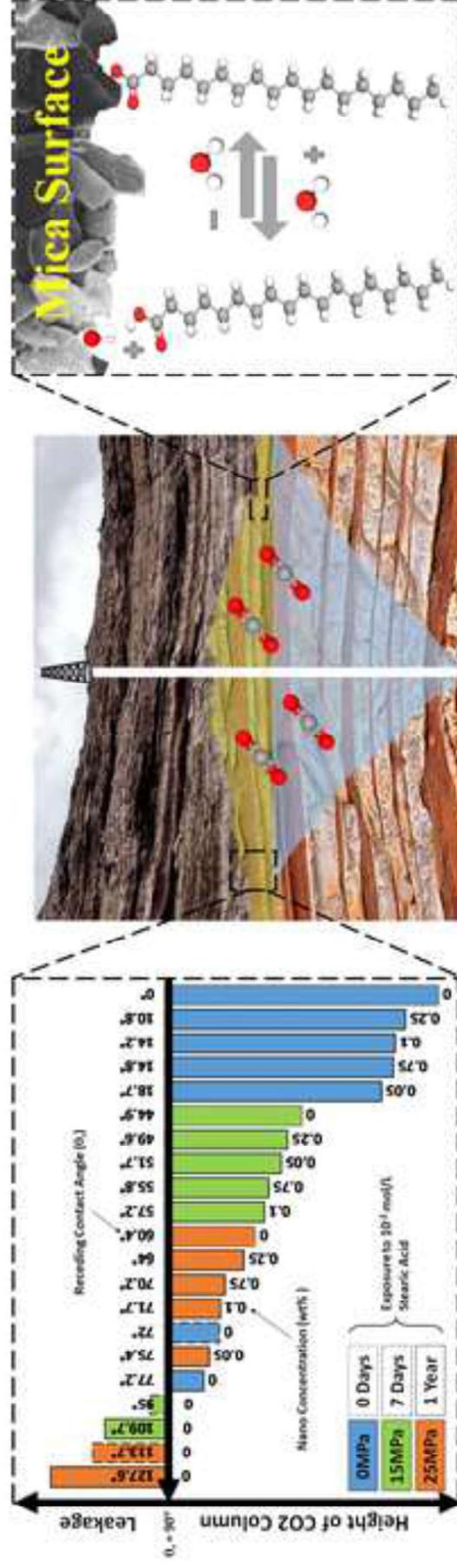
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Highlights:

- Wettability is directly or indirectly related to CO₂ structural trapping potential in cap-rock formation.
- Organic acids have a substantial effect on altering the wettability to CO₂-wet, thus reduced CO₂ trapping potential.
- Alumina Nanoparticles have significantly reversed the wettability to water-wet, thus increased CO₂ trapping potential.



1 **CO₂-Wettability Reversal of Cap-Rock by Alumina Nanofluid;** 2 **Implications for CO₂ Geo-Storage**

3

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25

26 **Abstract**

27 The usage of nanofluids is vast in different applications of nano-energy. These minute
28 nanoparticles can be used to alter the hydrophobicity into hydrophilicity for CO₂-brine-mineral
29 systems in the presence of organic acids. Nonetheless, the literature lacks the information for
30 the behavior of nanoparticles and its associated concentrations in the presence of organic acids
31 at the reservoir (high temperature and high pressure) conditions.

32 In this study, we have investigated that how different alkyl chain organic acids impact the
33 wettability of mica muscovite for different ageing times (7 days and one year) and how this
34 impact can be reversed by nanofluid priming at different concentrations. To do that, we have
35 used different organic acids with different alkyl chain lengths (hexanoic acid C₆, lauric acid
36 C₁₂, stearic acid C₁₈, and lignoceric acid C₂₄) at 10⁻² mol/L. We have also used different Al₂O₃
37 nanoparticle suspensions (0.05 wt%, 0.1 wt%, 0.25 wt% and 0.75 wt%) in deionised water.

38 When mica substrates were exposed to organic acids for a longer ageing time of 1 year, it has
39 lost its water-wetness rapidly at maximum. Whereas this effect was optimally reduced by 0.25
40 wt% of alumina nano-formulation and mechanical irreversible adsorption of alumina
41 nanoparticles was noted on mica substrates. This reversal of wettability may raise the
42 containment security and CO₂ structural trapping potential.

43

44 **Keywords:** Alumina Nano-particles, Organic Acids, Wettability, CO₂ Geo-storage, Deep
45 Saline Aquifers, Cap-rock Formation

46

47

48

49 **1. Introduction**

50 CO₂ geo-storage (CGS) sequestration is one of the auspicious technique used to immobilize
51 CO₂ in deep underground formations, thus contributing towards a green environment [1]. To
52 do that, CO₂ is captured from the environment and is stored in deep underground formations.
53 There are many methods that have been previously discussed for CO₂ capture [2-5], but storing
54 CO₂ deep underground formation is an only viable option for permanent CO₂ storage [6-8].
55 The projection of global CO₂ emissions in 2020 was 36.8 billion tons, which has now decreased
56 by 8% (2.94 billion tons) due to the novel COVID-19 pandemic. This pandemic has caused the
57 forced global industry to shut down, which has led to the sharpest fall of CO₂ emissions since
58 World War II [9]. However, future projection depicts that CO₂ emissions will gain momentum
59 towards 43.08 billion tons by 2050 [10]. The projected amount of CO₂ emissions is at an
60 alarming situation, and if its large portion is not stored in underground formations may lead to
61 accelerated global warming and irreversible climate change [1, 11].

62 Some of the prime candidates for CO₂ storage are deep saline aquifers, depleted petroleum oil
63 reservoirs, coal beds, basaltic formations, and organic shale reservoirs [1, 12, 13]. These
64 formations have vast storage capacities and can act as a barrier for CO₂ mobilization after its
65 storage to mitigate global warming [1, 6-8, 14]. After CO₂ injection in underground formations,
66 it is restrained by different mechanisms, which comprises of structural trapping (in cap-rock,
67 sandstone, and carbonate formations) [6, 15-18], residual or capillary trapping (in sandstone
68 and carbonate formations) [19-21], dissolution and mineral trapping (in basaltic, sandstone and
69 carbonate formations) [22-25], diffusion and adsorption trapping (in coal bed and organic shale
70 formations) [26-29].

71 In this context, wetting characteristics (wettability) of CO₂ geo-storage formation govern the
72 storage trapping capacities, containment security, production enhancement, and fluid dynamics

73 [30-34]. It is shown in previous studies that less water-wet conditions have shown the reduced
74 potential for CO₂ storage [17, 35-38]. However, there is a serious lack of data that can
75 comprehend complex wettability phenomena at realistic storage cap-rock environments. It is
76 proven in the literature that subsurface realistic storage environments are reductive or anoxic
77 in character, which contains highly diluted amounts of organic molecules [39-41]. These
78 minute organic concentrations are sufficient to alter wettability from water-wet to oil-wet [42-
79 44], thus, reducing storage capacities and containment security [7, 14].

80 In this perspective, alumina nanofluid suspensions can be a vital solution to diminish the effect
81 of reductive conditions in subsurface cap-rock storage formations. It is shown in the literature
82 that nanofluids are positively used for versatile applications of subsurface operations [45], like
83 chemical flooding [46], enhanced oil recovery [47-50], drilling [51-54], low salinity water
84 flooding [55-57], IFT reduction [58], and wettability alteration [8, 59, 60].

85 We have thus experimented CO₂/brine/Mica (muscovite, cap-rock proxy) contact angles
86 (advancing and receding) to investigate rock wetting characteristics (in the presence of organic
87 acids and alumina nanofluids) for determining CO₂ structural trapping capacities in cap-rock
88 formations. Shale caprock formations contains an abundant mineral called illite that has a
89 similar mineralogical structure like to mica (muscovite), which can be represented as a good
90 caprock proxy [15, 16, 61]. It is already proven in many previous studies that shale samples
91 already contain existing organics in them which has profound impact on wettability studies
92 [27, 62]. Therefore, we have used clean mica muscovite samples (as a proxy of caprock) to
93 obtain the correct thresholds for organic acids. It is also depicted in various previous wettability
94 studies, where mica has been used as a proxy of caprock formation [6, 15, 16]. Muscovite
95 chemically represented as $KAl_2(AlSi_3O_{10})(OH)_2$ [15, 16], and it is commonly present in mica
96 family with other members like, phlogopite, illite, and biotite, [63]. Further, existence of
97 muscovite is well proven in various sedimentary, metamorphic, and sedimentary formations

98 [64]. Previous studies have proven that these minerals wetting state is closely related with
99 structural trapping capacities [18, 19].

100 This information will shed light that how mica substrates aged in organic acids alter the
101 wettability to oil-wet, hence reduced CO₂ structural trapping capacity [6] and how these effects
102 can be reversed by ageing organic-aged mica substrates in alumina nanofluids, hence increased
103 CO₂ structural trapping capacity (similar trends were found in sandstone and carbonate
104 formations by using silica nanofluids) [8, 59]. It is vital for CGS projects to account for these
105 organic and alumina nanofluid thresholds for determining project feasibility and reduced
106 uncertainty.

107

108 **2. Experimental**

109 **2.1. Materials**

110 Hydrophilic alumina nanoparticles (Al₂O₃) were acquired from Sigma Aldrich (details are
111 given in Table 1) for formulating different nanofluid suspensions (0.05 wt%, 0.1 wt%, 0.25
112 wt%, and 0.75 wt%) in de-ionized (DI) water (Ultrapure of electrical conductivity = 0.02
113 mS/cm from David Gray). Pure mica crystals (dimensions = 15 mm x 20 mm x 3 mm, from
114 Ward's natural science) were used as the proxy of cap-rocks. Ten wt% brine solution was
115 prepared by mixing NaCl (purity ≥ 99.9 mol%, from Rowe Scientific) and deionized water as
116 a proxy of formation water for contact angle measurements.

117 It is proven in the literature that petroleum and gas reservoirs were the by-product of fossils,
118 which comprise organic acids (from C₄ to C₂₆) [65]. These organic acids are in abundant
119 quantity in depleted oil reservoirs and also occurs in deep saline aquifers due to organic
120 substance diagenesis and biodegradation of fossil mixtures where anaerobic situations flourish
121 [39-41]. In order to mimic the real reservoir situation for depicting the presence of organic

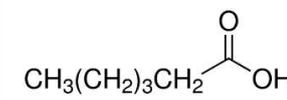
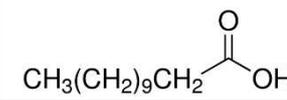
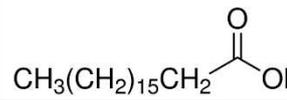
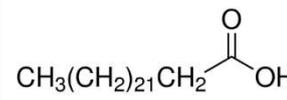
122 acids in storage formations, we have used four different organic acids (hexanoic acid, lauric
 123 acid, stearic acid, and lignoceric acid) ranging from C₆ to C₂₄ (purity ≥ 98 mol%, from Sigma
 124 Aldrich) for ageing mica substrates, details are explained in Table 1. The reason for choosing
 125 equal amount of difference in alkyl chain length of organic acids was to see the effect of alkyl
 126 chain length on wettability (explained in detail in section 3.1).

127 Drops of hydrochloric acid (HCl, the concentration of 37 vol%, from Sigma Aldrich) was
 128 introduced in 2 wt% NaCl solution for controlling its pH to 4 pKa and to ionize the mica
 129 substrates for speeding the ageing process of organic acids (details are given in section 2.2.2).
 130 For removal of contaminants from mica substrates before and after the ageing process, Ultra-
 131 pure nitrogen (purity = 99.999 wt%, from BOC, gas code-234) was utilized. Contact angle
 132 (advancing and receding) measurements were conducted in CO₂ (purity = 99.999 wt%, from
 133 BOC, gas code-082) atmosphere at high pressure and high temperature environments for
 134 simulating CO₂ geo-storage conditions. Methanol, toluene, acetone (purity ≥ 99.9 mol%, from
 135 Chemlab), and DI water were used to clean away pre-existing contaminations on mica
 136 substrates. n-Decane solution (purity ≥ 99.9 mol%, from Sigma Aldrich) was used to formulate
 137 various organic acid concentrations as the base liquid.

138

139 Table 1. Aluminum Oxide (Al₂O₃) nanoparticles and organic acids details.

Solubility in water	Insoluble
Molar weight (gm/mol)	101.96
Surface area (m ² /gm)	85-115
Particle Size	13 nm
Density (g/cm ³]	3.987
Purity (wt%)	≥ 99.8

Name of Organic Acid	pH (pKa)	Physical state	Formula	Molar mass (g/mol)	Number of C atoms	Chemical Structure
Hexanoic acid	4	Liquid	C ₆ H ₁₂ O ₂	116.158	6	
Lauric acid	5.3	solid	C ₁₂ H ₂₄ O ₂	200.318	12	
Stearic acid	6	solid	C ₁₈ H ₃₆ O ₂	284.4772	18	
Lignoceric acid	7.4	solid	C ₂₄ H ₄₈ O ₂	368.63	24	

140

141 2.2. The procedure of Mimicking Geo-Storage Environments for Mica Surfaces

142 2.2.1. Initial Cleaning Procedure of Pristine Mica

143 Pristine mica substrates contain impurities which may affect the experimental results. In order
 144 to avoid this, it is pertinent to clean pure mica substrates as they are received. We thus cleaned
 145 pure mica samples with deionized water and acetone to eliminate any in-organic or organic
 146 surface impurities. This procedure was followed by blow of ultra-pure nitrogen on the
 147 substrate's surfaces; as well as a vacuum oven was used for the drying process (80 °C, time =
 148 120 minutes) for removal of any remaining fluid after preliminary cleaning. Afterward, air
 149 plasma (Diemer Yocto instrument) was used to treat the mica substrates for 20 minutes to
 150 eliminate any residual fragments after the drying procedure [66].

151

152 2.2.2. Simulating Mica Substrates as Geo-Storage Formation

153 CO₂ storage formations are anoxic in character which contains organic impurities that lie in the
 154 exposure of formation waters (note this exposure can be for millions of years). Therefore, it is
 155 pertinent to simulate a similar mineralogical situation in-order to fully comprehend the

156 underground physiognomies for the contact angle experiments [67, 68]. There is a serious lack
157 of data in the literature for comprehending the effects of organics on various types of storage
158 formations at various geophysical conditions. However, to do this, silanes were utilized earlier
159 for modifying the hydrophilic surfaces [69], but real reductive conditions for storage
160 formations do not contain silanes (note silanes are highly reactive in reductive conditions). It
161 is previously proven that deep saline aquifers contain enough volume of organics to alter their
162 wettability from water-wet to oil-wet [6, 39-41], and their presence is possible due to organic
163 substance diagenesis and biodegradation of fossil mixtures where anaerobic situations flourish
164 [70, 71]. Therefore, it is more realistic to use organic acids for modifying the wettability of the
165 rock matrix, instead of using silanes at laboratory scale [42, 59]. This modification of mineral
166 hydrophobicity happens due to chemical bonding between hydroxyl groups of mineralogical
167 surfaces and organic esterification [6, 7, 14, 59]. Thus we followed the subsequent approach
168 [67, 68, 72, 73].

169 Initially, NaCl brine (2 wt%) was formulated with deionized water, and drops of aqueous HCl
170 were added for maintaining the pH at four pKa. Afterward, pristine mica substrates were aged
171 with a solid/liquid ratio of 1:5 in pH-controlled two wt% NaCl brine. This process ionizes the
172 pristine mica substrates in order to increase the adsorption rate of organic acids for mimicking
173 the millions of year's formation water exposure [6, 7, 14, 43, 59, 74]. Thereafter, a blow of
174 ultra-pure nitrogen was used to remove the brine solution from sample surfaces to elude
175 impurities. Consequently, ionized mica substrates were aged in different organic acid/n-decane
176 solutions of prescribed concentration of 10^{-2} mol/L (note four mica substrates were aged in
177 each organic acid for each nanofluid suspension). Our mica substrates remained in organic
178 acid/n-decane solutions for seven days (initial contact angle measurement as a function of
179 ageing) and one year (subsequent contact angle measurement as a function of ageing). This
180 variation in ageing is due to provide enough interval for mimicking formation performance to

181 accurately comprehend wettability characteristics [8, 67, 68, 75]. However, ageing with 7 days
182 and one year does not show a significant difference in contact angle measurements (explained
183 in section 3.1), this is due to the optimum ageing effect was already achieved in 7 days and
184 longer interval ageing of one year have only caused an insignificant forced ageing resulting in
185 less contact angle differences between 7 days aged samples and one year aged samples.

186 In-order to contain exact organic acid concentration (10^{-2} mol/L in our case), acrylic containers
187 with tightly sealed caps were used for the ageing process (especially useful in 1-year ageing)
188 to elude n-decane evaporation. This evaporation can cause a change in organic concentration
189 and subsequent error in comprehending the exact wettability threshold. Afterward, acrylic
190 containers were stored in a closed environment (fume hood) to elude impurities for a year.

191

192 **2.3. Ageing of Organic Aged Mica Substrates in Alumina Nanofluids**

193 **2.3.1. Formulation of Alumina Nano-Fluid**

194 Various concentrations (0.05 wt%, 0.1 wt%, 0.25 wt%, and 0.75 wt%) of alumina nano-
195 formulations were formulated (nanofluid formulation was done by sonication method) in order
196 to comprehend their competence for wettability reversal of CO₂/brine/mica systems. The
197 prescribed weight of Al₂O₃ nano-particles (for various concentrations) was added to deionized
198 water and sonicated (Ultrasonic homogenizer, Frequency 20 kHz, Sonics and Materials
199 Incorporation, USA) for 15 minutes (note that homogeneous stirring is not possible with a
200 magnetic stirrer) [76]. This process is conducted with a micro tip (titanium material) of 9.5 mm
201 diameter with a sonication amplitude of (40 %) and energy of (9500 Joules).

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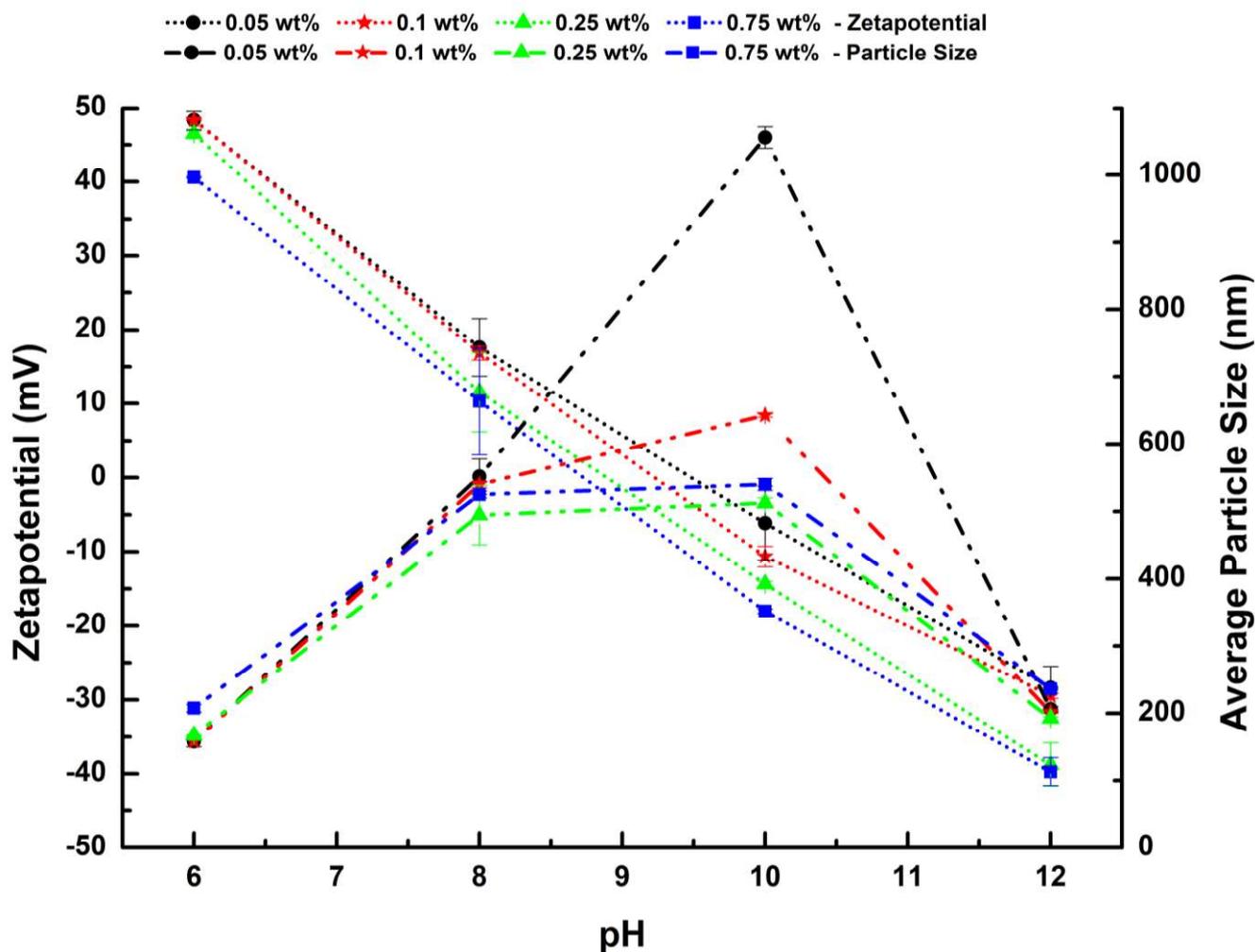
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204 **2.3.2. Alumina Nano-Fluid's Stability**

205 It is crucial to determine the stability of nano-formulations, which is a pertinent aspect to
206 comprehend its competence for reversal of wettability [8]. To do this, nanofluid electrical
207 stability phase behavior, i.e., Zeta potential (mV) and average particle size (nm) results for
208 alumina nanoparticles were measured from pH level of 6 to 12 (note same range has been noted
209 in deep saline aquifers [77]. Zeta sizer Nano (Malvern, ZS ZEN 3600, Malvern Instruments,
210 UK) was employed for each nanofluid concentration (0.05 to 0.75 wt%), while pH was
211 measured using a pH meter (resolution 0.01 pH, Ohaus, Australia).

212 Lower Zeta potential values (+ 10 to -10 mV) of the nano-formulation are incipiently
213 electrically stable and are prone to aggregate quicker than as compared to higher values (i.e.,
214 Absolute zeta potential > 35 mV) [78, 79]. Furthermore, the repulsive forces (positive or
215 negative charged) are significantly decreased, resulting in acceleration of the nanoparticle
216 coagulation process that yields sedimentation in the presence of the electrolyte. Cationic and
217 anionic surfactants can alter the electrical stability (i.e., zeta potential) of the particles (in the
218 present case, alumina nanoparticles) by coating particle's surface with their counter-ion head
219 group, hence resulting in supercharged particles [80, 81].

220



221

222 Figure 1. Zeta potential and average particle size measurements of various alumina nano-
 223 particles concentration (0.05, 0.1, 0.25 and 0.75 wt%) as a function of pH.

224

225 The zeta potential measurement depicts that the alumina nanoparticles have an isoelectric point
 226 (IEP, i.e., $\zeta=0$) at a pH level of 8 to 10 in DI water, as shown in Figure 1. It is noteworthy to
 227 mention here that, as the weight percentage is increased from 0.05 to 0.75, a decline in zeta-
 228 potential on the positive side is observed (e.g., at pH = 6), however when we observe on the
 229 highly alkaline side (i.e., pH = 12) an inverse phenomenon is observed. This change of behavior
 230 is due to OH⁻ ions added to the nano-formulation. As more alumina nanoparticle adsorptive
 231 sites at more significant loading (i.e., 0.75 wt%), the more OH⁻ gets attached to the nanoparticle

232 surface, hence resulting in higher dispersion stability [79]. Generally, the data shows that with
233 increasing concentration of nanofluids, the average particle size (d_{av}) has increased (up to 10
234 pH). However, upon a further increase in pH, the d_{av} has decreased owing to the enhancement
235 of electrostatic repulsion due to the absorption of OH^- ions, also resulting in higher zeta
236 potential value (on the positive side).

237 Alumina nanofluid is an excellent candidate for muscovite owing to its broad spectrum of zeta
238 potential (i.e., from 48 to -40 mV) and its associated stability across a wide pH range (i.e., from
239 6 to 12 pH), as can be seen in Figure 1. The absolute difference in surface roughness owing to
240 the presence of nanoparticles can be examined quantitatively, as can be seen in Table 2. It is
241 hypothesized that alumina nanoparticles are chemisorbed onto the muscovite surface due to its
242 surface activity, as can be seen in Figure 1, for which the presence of OH^- charges plays a vital
243 role in interacting with positively charge mica surface.

244

245 **2.3.3. Ageing Procedure of Organic Aged Mica Substrate's in Alumina Nano-Fluids**

246 It is proven in the literature that nanofluids have considerable competence in comprehending
247 wettability reversal [8, 47-49]. To do this, organic/n-decane aged mica substrates were
248 vertically submerged (note vertical submerging of mica samples reduces gravitational
249 deposition of nanoparticles) for 5 hours in various alumina nano-formulations (0.05 wt%, 0.1
250 wt%, 0.25 wt%, and 0.75 wt%) at elevated temperature 323 K and atmospheric pressure to
251 form nano-modified mica surfaces. This ageing is obtained with a solid/nanofluid ratio of 1:5
252 (1 gram of mica substrate for 5 grams of nano-formulation) for delivering the same coverage
253 for each mica surface [8, 59].

254 Mechanistically, the nanofluid ageing process provides a very strong chemical reaction by the
255 ageing of the hydroxyl group of nano-formulation with a hydroxyl group of mica substrate

256 (note this chemical reaction is the reason of irreversible adsorption of nanoparticles on mica
257 substrates). This irreversible adsorption of nanoparticles on organic/n-decane aged mica
258 substrates is responsible for shifting the hydrophobicity into hydrophilicity. Previously, quartz
259 substrates aged in organic acids were placed in negatively charges SiO₂ nanoparticle solutions,
260 which has shown similar significant wettability reversal to water-wet [8, 59].

261 Overall, the hydroxyl group of mica substrates is esterified with organic acids in a condensation
262 reaction (shifting the wettability to CO₂-wet), and nanoparticles cause irreversible adsorption
263 on organic/n-decane aged mica substrates in a chemical reaction (reversing the wettability to
264 weakly water-wet).

265

266 **2.4. Characterization of Nano-aged, Organic-aged, and Pristine Mica substrates**

267 **2.4.1. Surface Chemistry**

268 Surface elemental analysis (in wt%) of each mica sample was carried out using a Field
269 Emission Scanning Electron Microscope (FESEM) from Oxford Instruments via energy
270 dispersive spectroscopy (EDS) technique. These measurements were done on all pure,
271 organic/n-decane aged and nano-aged mica substrates. It is clear from our results that the
272 ageing of organic acids and nano-formulations have a significant effect on the surface
273 elemental analysis (average elemental values of pure, organic-aged, and nano-aged mica
274 substrates are depicted in Table 2). A significant average increase in surface alumina
275 concentration (+19.1 wt% Al, for Hexanoic-aged surface, +17.4 wt% Al, for Lauric-aged
276 surface, +21.4 wt% Al, for Stearic-aged surface, and +18.5 wt% Al, for Lignoceric-aged
277 surface) due to adsorption of nanoparticles [8] and in surface carbon concentration (+6.1 wt%
278 C, for Lignoceric Acid, +4.5 wt% C, for Stearic Acid, +3.8 wt% C, for Lauric Acid and +2.6
279 wt% C, for Hexanoic Acid) due to chemisorption of carboxylic acids was noted on mica

280 substrates [6, 72]. These values are the average of 5 data points for each, pure, organic-aged,
 281 and nano-aged mica substrates (A detailed surface elemental analysis is given in Table S1).
 282 Further, high magnification scanning electron micrographs (SEM) were acquired on pure and
 283 nano-aged mica substrates for depicting the irreversible adsorption of nano-formulations
 284 (Figure 2, it is clearly shown that 0.25 wt% Alumina nano-formulation have uniformly
 285 adsorbed on mica substrate). Such uniform adsorption of nano-formulation is responsible for
 286 wettability reversal from a strongly hydrophobic surface to a weakly hydrophilic surface. "±"
 287 shows standard deviation in all mica substrates tested in that particular range (Table 2).

288

289 Table 2. Average surface elemental analysis of pristine, organic-aged, and nano-aged mica
 290 substrates*.

Organic Acids	Average elemental values of pristine mica					Average elemental values after organic-aged mica (10 ⁻² mol/L)					Average elemental values after nano-aged mica (0.05 wt%, 0.1 wt%, 0.25 wt% and 0.75 wt%)				
	wt% Al	wt% Si	wt% C	wt% O	wt% K	wt% Al	wt% Si	wt% C	wt% O	wt% K	wt% Al	wt% Si	wt% C	wt% O	wt% K
Hexanoic acid	19.4 ± 0.8	21.9 ± 1.0	3.2 ± 0.4	46.9 ± 0.5	8.7 ± 0.6	18.7 ± 1.1	20.5 ± 1.7	5.7 ± 0.4	47.8 ± 3.0	7.4 ± 0.7	37.8 ± 7.4	13.2 ± 3.8	4.8 ± 0.5	38.4 ± 3.0	5.8 ± 0.5
Lauric acid	19.3 ± 0.9	22.3 ± 0.7	2.8 ± 0.3	47.1 ± 1.3	8.5 ± 0.6	18.6 ± 0.6	19.9 ± 1.4	6.6 ± 0.3	47.6 ± 1.0	7.4 ± 1.0	35.9 ± 8.9	13.3 ± 3.3	5.3 ± 0.4	39.9 ± 5.5	5.6 ± 0.6
Stearic acid	19.5 ± 0.8	22.2 ± 1.5	3.2 ± 0.7	46.6 ± 0.8	8.6 ± 0.8	18.5 ± 0.9	20.5 ± 1.5	7.7 ± 0.3	45.8 ± 1.5	7.5 ± 0.7	40.0 ± 8.6	11.7 ± 3.2	5.9 ± 0.2	36.5 ± 5.2	6.0 ± 0.5
Lignoceric acid	19.4 ± 0.7	22.4 ± 1.2	3.1 ± 0.6	46.4 ± 1.2	8.8 ± 0.5	18.8 ± 1.1	20.3 ± 0.5	9.1 ± 0.2	44.6 ± 1.8	7.2 ± 0.7	37.3 ± 8.9	13.2 ± 2.6	6.8 ± 0.4	37.1 ± 5.4	5.7 ± 0.6

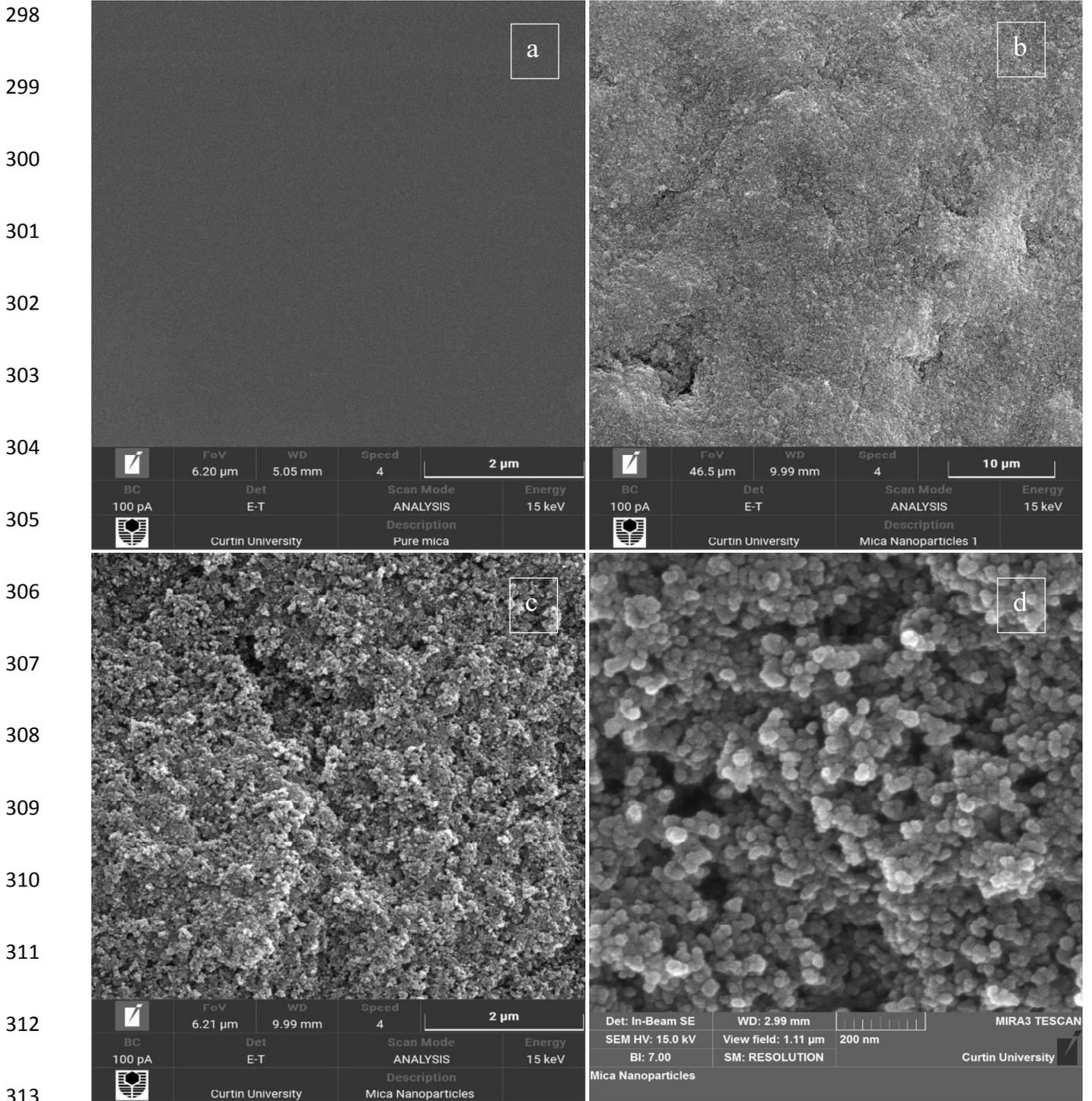
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292 * Average of 5 data points of each, pure, organic-aged, and nano-aged mica substrates of four
 293 different organic acids (hexanoic, lauric, stearic, and lignoceric acid) and four different nano-
 294 formulations (0.05 wt%, 0.1 wt%, 0.25 wt%, and 0.75 wt%).

295

296

297



314 Figure 2. Different magnifications of SEM Micrographs (a) Pristine mica at a magnification of
 315 2 μm (b) 0.25 wt% Alumina nano-aged mica surface at a magnification of 10 μm (c) 0.25 wt%
 316 Alumina nano-aged mica surface at a magnification of 2 μm (d) 0.25 wt% Alumina nano-aged
 317 mica surface at a magnification of 200 nm.

318 **2.4.2. Surface Roughness**

319 Surface roughness has a substantial impact on contact angle measurements, hence wettability
320 [82]. Therefore, all substrates (pristine mica, organic-aged mica, and nano-aged mica) were
321 characterized by conducting surface topography (Atomic force microscopy by Flex-Axiom,
322 Controller C3000 AFM instrument from Nano-surf) measurements. Topographic experiments
323 have revealed that pristine mica substrates have a very smooth surface (route mean square,
324 RM), ranging from 1 to 2 nm [6]. Further, the topography of organic-aged and nano-aged mica
325 substrates have also depicted homogenous and smooth surfaces with an average root mean
326 square ranging from 200 to 460 nm (for organic-aged mica) [6] and 340 to 620 nm (for nano-
327 aged mica). It is proven in the literature that surfaces with average route mean square roughness
328 for more than 1 μm has a considerable effect on wettability measurements [83]. Hence, all mica
329 substrate's surface roughness resulted in less than 1 μm value; therefore, roughness has no
330 impact on contact angle experiments [84].

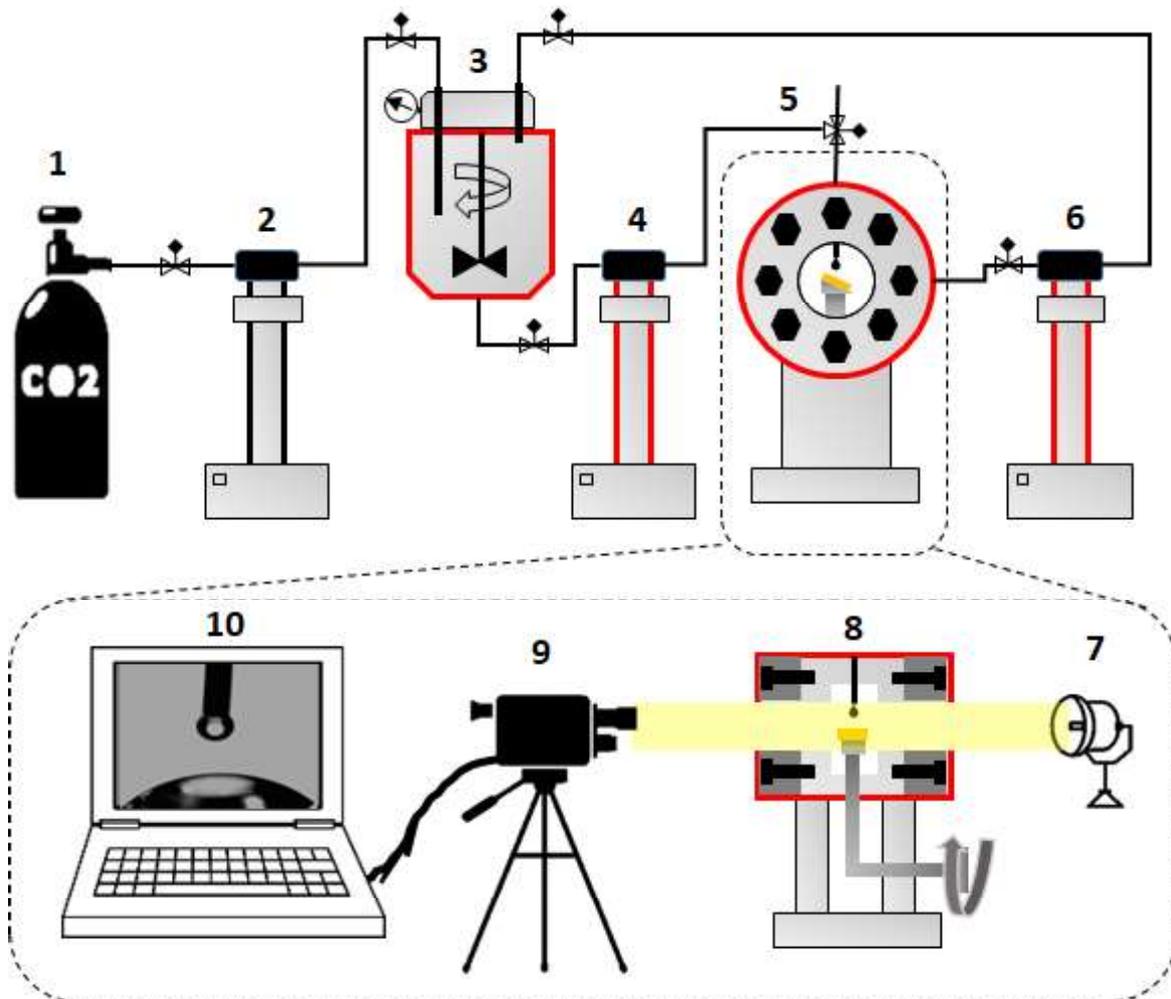
331

332 **2.5. Contact Angle Experiments**

333 There are several methods to quantify wettability behavior, like, capillary pressure and relative
334 permeability curve, nuclear magnetic resonance (NMR) technique, spinning drop technique,
335 Wilhelmy balance, and capillary rise method, Amott-Harvey index, and US Bureau of Mines
336 (USBM) core flood method. All these methods can give indirect wettability calculation,
337 whereas, contact angle method is the only method that gives direct quantitative wettability
338 calculation in a given rock fluid system [8, 16-18, 59]. This method contains a tilted surface
339 technique for the calculation of advancing (θ_a) and receding (θ_r) contact angles [85].

340 In this technique (Figure 3), a flat mineral sample surface (mica in our case) is placed inside
341 the high pressure and high temperature cell (HPHT, Hastelloy material made) on a tilted plate.

342 For a relative rock fluid system, it is pertinent to obtain thermodynamically equilibrated brine
343 solution with CO₂ and mica substrates. Mica is a reactive surface which can react in acidic
344 environments due to the presence of CO₂ [86, 87], causing errors in wettability experiments.
345 Therefore, brine solution is introduced in Parr reactor mixed with CO₂ and mica substrates at
346 desired experimental (high pressure and high temperature) conditions to obtain thermodynamic
347 equilibration before the contact angle measurements [88]. Thereafter, gas (CO₂ in our case) is
348 introduced into the HPHT cell at desired reservoir conditions (323 K, 0.1 MPa, 15 MPa, and
349 25 MPa in our case) controlled by a high precision syringe pump (Teledyne ISCO, Model D-
350 500, pressure accuracy of 0.1%). Afterward, a drop of thermodynamically equilibrated brine
351 (average drop size was $4.5 \mu\text{L} \pm 0.75 \mu\text{L}$) is dispensed (controlled by another high precision
352 syringe pump) on the tilted mineral surface. This process is video recorded for determining
353 advancing (θ_a) and receding (θ_r) contact angles at the front and back edges of the brine droplet
354 [43]. The schematic of the HPHT contact angle system is depicted in (Figure 3). A further
355 detailed description of this process is given in our previous articles [6-8, 14].



356

357 Figure 3. Schematic of HPHT contact angle system (1) CO₂ bottle (2) ISCO syringe pump for
 358 CO₂ (3) High-pressure Parr reactor for live brine formulation (4) ISCO syringe pump for live
 359 brine (5) HPHT Hastelloy cell with tilted plate housing, front view (6) ISCO syringe pump for
 360 wet CO₂ (7) Light projection (8) HPHT Hastelloy cell with tilted plate housing, side view (9)
 361 High-resolution video camera (10) ImageJ software for interpretation.

362

363 3. Results and Discussion

364 It is pertinent to comprehend the quantitative assessment of CO₂-wettability for cap-rock
 365 formations (mica in our case). This assessment is crucial for determining the CO₂ drive and its
 366 distribution across the storage formation [24, 35], storage capacities [16, 17], flow properties

367 [18, 89], and containment security [90]. In this regard, capillary outflow can happen when the
368 receding contact angle of brine (10 wt% NaCl in our case) is greater than 90° (this phenomenon
369 represents CO_2 injection into the storage formation while displacing water), thus reduced
370 structural trapping capacity [31]. Whereas, wettability have no impact on primary drainage
371 when the advancing contact angle of brine is less than 50° (this phenomenon represents water
372 injection into the storage formation to capillary trap CO_2) [38, 61]. It is also proven that
373 dissolution trapping is substantially impacted by CO_2 -wettability alteration [22, 23].

374 It is clear from our contact angle measurements that hydrophilic mica surfaces were greatly
375 affected by the presence of organic acids and rapidly lost their water-wetness. This wettability
376 alteration can greatly affect the structural trapping capacities of mica cap-rock, whereas this
377 wettability shift can be reversed by alumina nanofluid treatment. Initially, pristine mica
378 substrates were tested at atmospheric pressure and elevated temperature (323 K), which
379 resulted in a completely hydrophilic state with $\theta = 0^\circ$ [6, 15, 16]. Whereas, reservoir conditions
380 (323 K, 15 MPa and 25 MPa) have depicted weakly water-wet state ($\theta_a = 50.2^\circ \pm 3^\circ$ and $\theta_r =$
381 $44.9^\circ \pm 3^\circ$ at 15 MPa; and $\theta_a = 65.1^\circ \pm 5^\circ$ and $\theta_r = 60.4^\circ \pm 5^\circ$ at 25 MPa) [6, 15, 16].

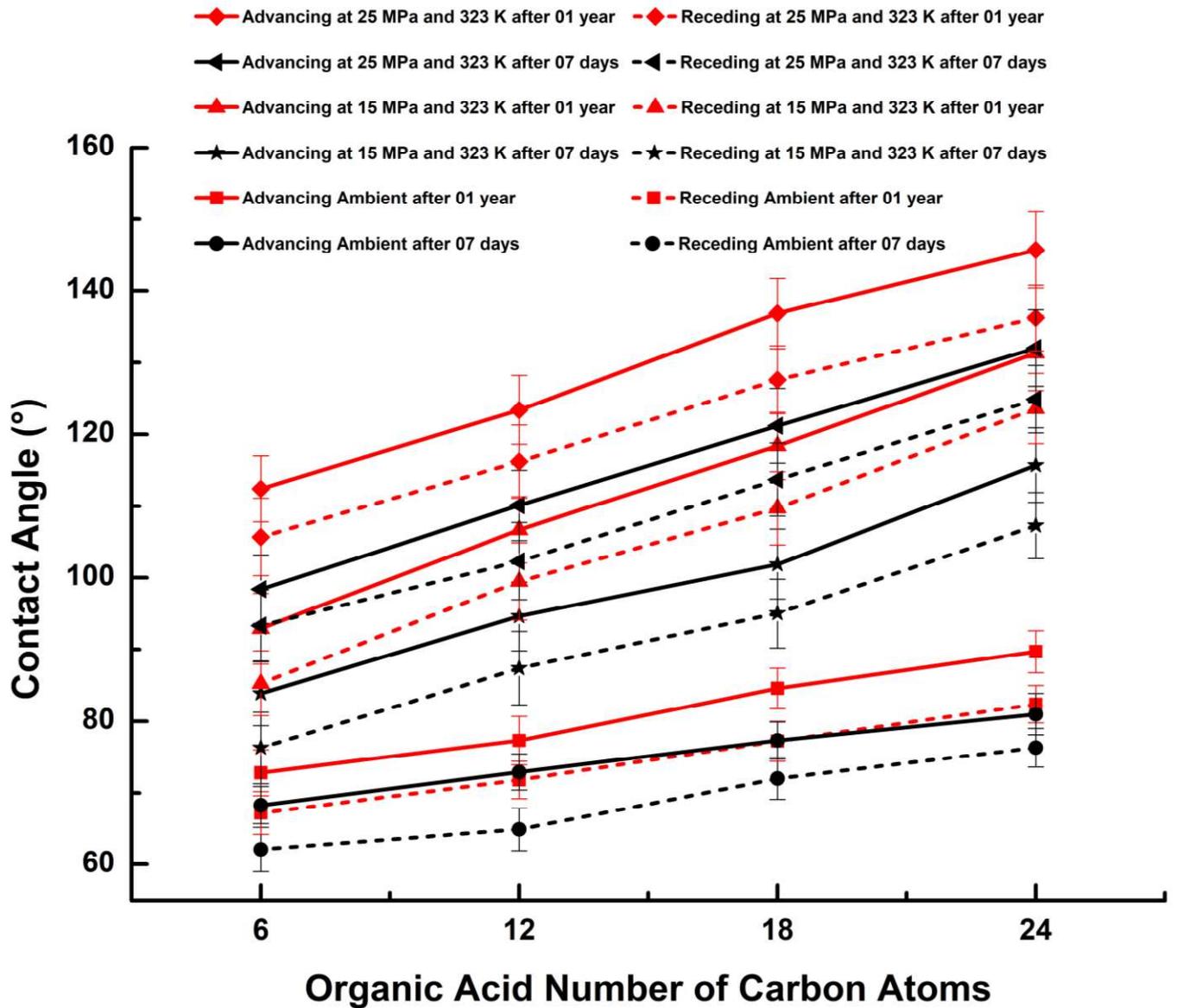
382

383 **3.1. Effect of Organic Acids as a function of Ageing Time, alkyl chain length and** 384 **Pressure on CO_2 Geo-Storage Capacities**

385 CO_2 -wettability of cap-rock formation is directly or indirectly related to CO_2 structural trapping
386 capacity. In ideal situations, wettability drives the ability of CO_2 to displace the water and
387 induce below the cap-rock formation. This happens when CO_2 move up-ward (due to buoyancy
388 factor) and apply force on solid cap-rock formation; thus, structurally trapping itself. However,
389 in real situations, organic fatty acids are present in the storage formation, which can
390 significantly affect the wettability, thus, causing capillary leakage [6, 40, 41].

391 Our results have shown that at a constant organic acid (hexanoic acid C₆, lauric acid C₁₂, stearic
392 acid C₁₈, and lignoceric acid C₂₄) concentration of 10⁻² mol/L with different ageing times (7
393 days and one year), mica has greatly lost its water-wetness. For instance, at a constant pressure
394 of 15 MPa and 323 K for lignoceric acid with one-year ageing time, mica/CO₂/brine contact
395 angles were strongly hydrophobic ($\theta_a = 131.3^\circ$ and $\theta_r = 123.6^\circ$) and at similar conditions, with
396 seven days ageing time contact angle values were ($\theta_a = 115.7^\circ$ and $\theta_r = 107.3^\circ$) compared to
397 pristine mica ($\theta_a = 50.2^\circ$ and $\theta_r = 44.9^\circ$), which may have serious consequences in structural
398 trapping capacities due to prolonged exposure of organic acids (Figure 4).

399 However, irrespective of ageing time, longer alkyl chain (higher number of carbon atoms)
400 organic acids have a more negative effect on structural trapping (increased contact angle
401 values) compared to shorter alkyl chains (fewer number of carbon atoms) organic acids.
402 Therefore, mica substrates aged in lignoceric acid (C₂₄) have shown higher contact angle values
403 compared to mica substrates aged in hexanoic acid (C₆). For instance, at 25 MPa and 323 K,
404 with an ageing time of one year, of all organic acids (hexanoic acid C₆, lauric acid C₁₂, stearic
405 acid C₁₈, and lignoceric acid C₂₄), receding contact angle values were, $\theta_r = 105.7^\circ$, 116.2° ,
406 127.6° , and 136.2° , respectively (Figure 4). Such higher contact angle values can significantly
407 reduce structural trapping capacities (note structural leakage is possible at $\theta_r > 90^\circ$) [6, 16-18].



408

409 Figure 4. Mica/CO₂/Brine advancing and receding contact angles as a function of ageing time,
 410 different organic acids, and pressure.

411

412 Further, irrespective of ageing time and type of organic acid, higher pressure (deep saline
 413 aquifers > 15 MPa) may cause higher threat (increasing contact angle values) for structural
 414 trapping capacities due to an increase in intermolecular forces between solid (mica in our case)
 415 and gas (CO₂ in our case) with increasing pressure [91]. For example, mica substrates aged in
 416 stearic acid at atmospheric pressure and elevated temperature (323 K) with seven days ageing

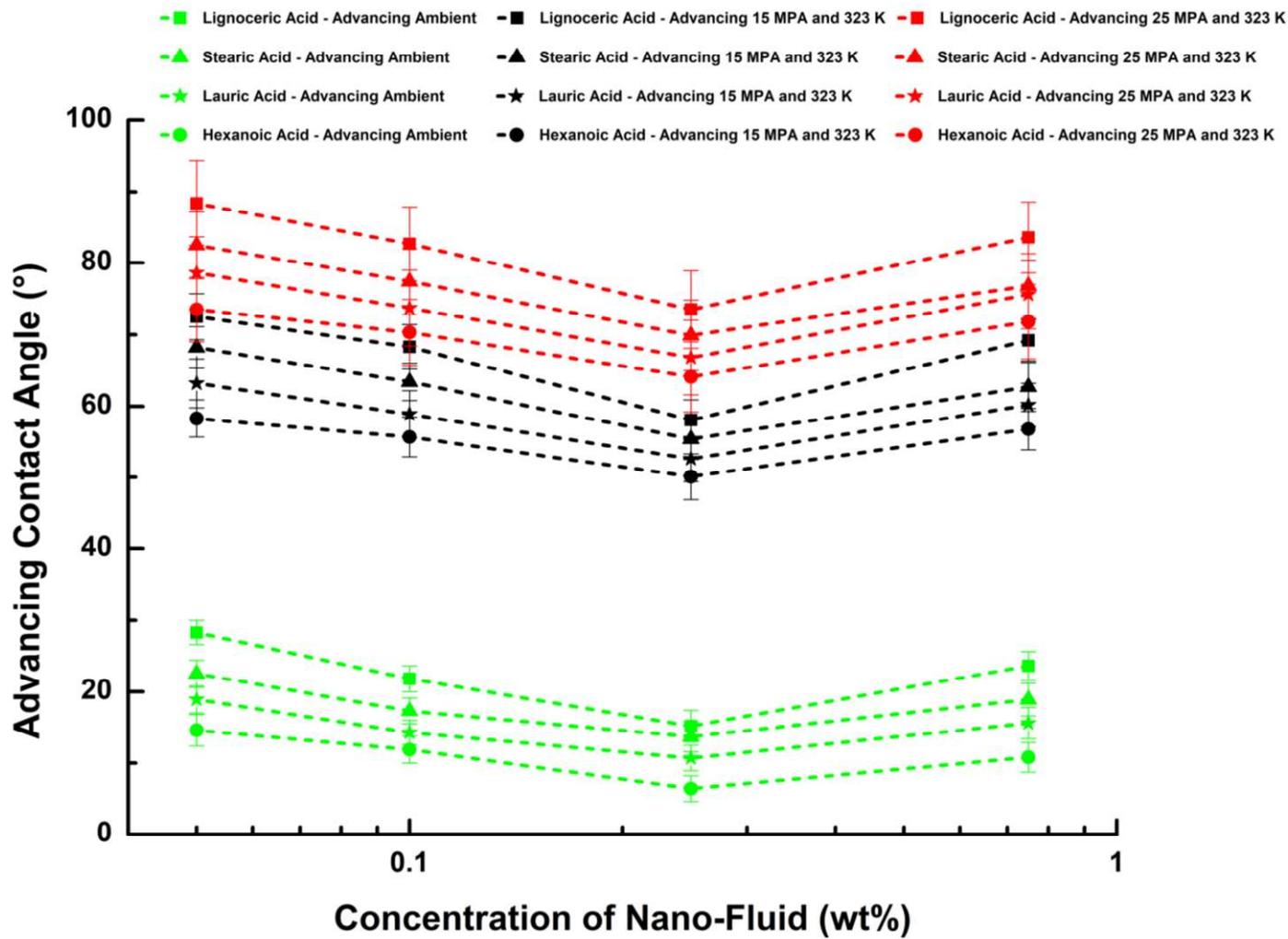
417 time have shown weakly intermediate-wet state $\theta_a = 72.9^\circ$ and $\theta_r = 64.9^\circ$, whereas, at similar
418 conditions with higher pressure of 25 MPa, mica/CO₂/brine contact angle values turned in CO₂-
419 wet $\theta_a = 110.1^\circ$ and $\theta_r = 102.3^\circ$ (Figure 4). Hence, such minute concentrations of organic acids
420 and higher pressures are always present in storage formations; lower structural trapping
421 capacities may be encountered than estimated [7, 39, 70, 71].

422

423 **3.2. Effect of Nano-Fluids on CO₂ Geo-Storage Capacities**

424 It is shown in the literature that various parameters like temperature, pressure, brine salinity,
425 and organic acids have a substantial impact on CO₂ geo-storage [15, 16, 89]. However, there
426 is a serious lack of data for comprehending the effect of nanoparticles (type and concentration)
427 at HPHT reservoir conditions for CO₂-wettability, hence CO₂-storage capacities. To do that,
428 we have used different concentrations of alumina nanoparticles (0.05 wt%, 0.1 wt%, 0.25 wt%,
429 and 0.75 wt%) in deionized water for the reversal of hydrophobic mica surfaces. Organic-aged
430 mica substrates were aged vertically in these nanofluid concentrations for 5 hours, followed by
431 contact angle measurements at reservoir conditions (323 K, 0.1 MPa, 15 MPa, and 25 MPa in
432 our case). This nano-treatment of mica substrates is achieved by the adsorption efficiency of
433 alumina nanoparticles and their associated concentrations. It is depicted from our results (figure
434 5 and 6, advancing and receding contact angles with alumina nanofluid) that the wettability of
435 organic-aged CO₂-wet mica substrates have substantially reduced by alumina nanofluid ageing
436 (note the degree of wettability reversal was different for different alumina nanofluid
437 concentrations) [92].

438



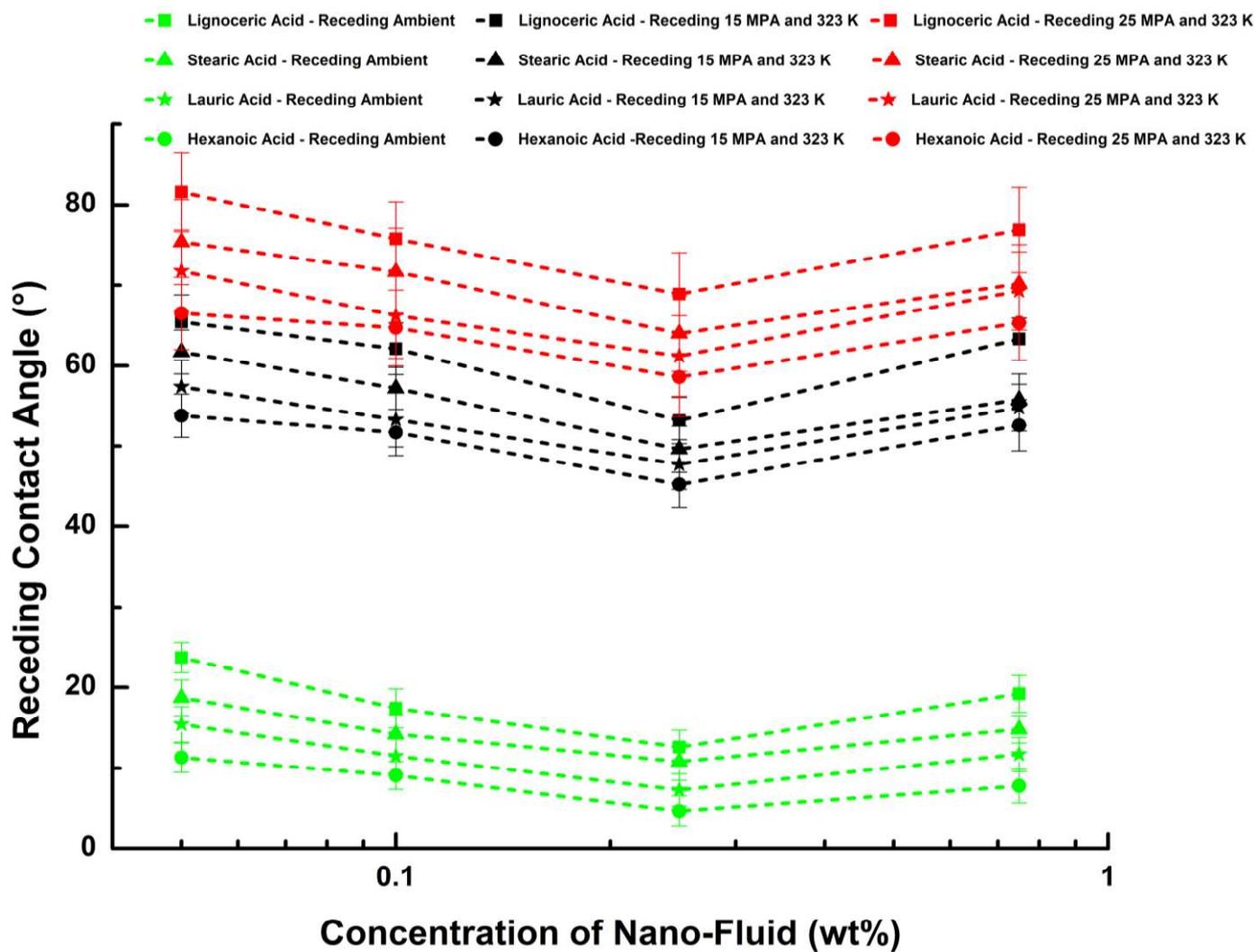
439

440 Figure 5. Mica/CO₂/Brine contact angles (advancing) as a function of pressure and different
 441 alumina nanofluid concentrations.

442

443 For instance, with lignoceric-aged mica substrates at 323 K and 25 MPa for 0.05 wt%, 0.1 wt%,
 444 0.25 wt%, and 0.75 wt% nanofluid alumina concentrations, brine receding contact angles were
 445 $\theta_r = 81.6^\circ$, 75.8° , 68.9° , and 76.9° , respectively (note structural trapping limit is $\leq 90^\circ$), which
 446 has reduced from $\theta_r = 136.2^\circ$ for lignoceric-aged mica substrates at 323 K and 25 MPa for one-
 447 year ageing. It can be seen from the acquired data that each concentration of alumina nanofluid
 448 is setting the same trend for other organic-aged mica substrates; however, 0.25 wt% alumina

449 nanofluid concentration has shown the optimum reversal of wettability as compare to other
 450 alumina nanofluid concentrations tested.



451

452 Figure 6. Mica/CO₂/Brine contact angles (receding) as a function of pressure and different
 453 alumina nanofluid concentrations.

454

455 **4. Implications**

456 The conducted contact angle experiments have depicted that the existence of organic acids into
 457 CO₂ geo-storage formations (mica in our case) is responsible for shifting the wettability to
 458 more CO₂-wet, thus, reduced structural trapping capacities [6, 18]. Therefore, optimum

459 alumina nanofluid concentration (0.25 wt% in deionized water) is recommended for the
460 reversal of wettability from hydrophobicity to hydrophilicity, hence, increased structural
461 trapping capacities, as depicted in similar studies for applications of SiO₂ nanoparticles in
462 quartz formation [8]. This can be achieved by following two processes.

- 463 1. Initially, by injecting optimum nanofluid suspension (0.25 wt% in deionized water) in
464 CO₂-geo storage formation to alter the wettability, which is CO₂-wet due to the
465 presence of organic acids (hence less CO₂ structural trapping potential). This will
466 change its wetting characteristics to Intermediate-wet (hence greater CO₂ structural
467 trapping potential).
- 468 2. Finally, when wetting characteristics are shifted towards intermediate-wet, CO₂ will be
469 injected at above supercritical pressure, which will exert the pressure on wetting phase
470 (formation water) and free CO₂ will move upward due to the buoyancy force and will
471 accumulate below the impermeable layer, where it will be structurally trapped below
472 the caprock formation.

473 Consequently, alumina nanofluid injection may play an integral part as part of CO₂ geo-storage
474 feasibility. Therefore, CO₂ geo-storage projects should account for the strategies where leakage
475 risk can be minimized.

476

477 **5. Conclusion**

478 Various factors such as formation heterogeneity, brine salinity, temperature, surface roughness,
479 and pressure are basic fundamentals that drive CO₂-wettability of storage formations [14, 16,
480 55-57, 83, 89, 93, 94]. Nonetheless, organic acids' existence in storage formation is well-
481 proven, which may significantly reduce structural trapping capacities [6, 7, 14, 43], and this
482 may be reduced with different concentrations of nano-fluids in nano-energy applications [8,

483 47-49, 59]. There is a serious lack of data in the literature that can account for the CO₂-
484 wettability in the existence of organic acids for mitigating project achievability, technical
485 injection problems, containment security, and structural trapping feasibility in cap-rock
486 formation [7, 12, 14, 17, 18, 35, 43]. In contrast, many studies of storage formations have
487 already shown the presence of organics at a larger scale [7, 39-42, 73, 74].

488 In order to properly address this issue. We have used four organic acids (hexanoic, lauric,
489 stearic, and lignoceric acid) at a fixed organic acid/n-decane concentration of 10⁻² mol/L with
490 different exposure durations (7 days and one year) for ageing pure mica substrates. Thereafter,
491 different concentrations (0.05 wt%, 0.1 wt%, 0.25 wt%, and 0.75 wt%) of alumina nanofluids
492 were formulated with deionized water for ageing (5-hour duration) organic-aged mica
493 substrates. Contact angle measurements (advancing and receding) were conducted at various
494 storage conditions (323 K, 0.1 MPa, 15 MPa, and 25 MPa) in-order to fully comprehend the
495 wetting characteristics of organic-aged and nano-aged mica substrates, thus, determining
496 structural trapping limits (leakage can occur at > 90°) in the presence of organic acids and
497 nanofluids.

498 Our results depict that pristine mica substrates were weakly water-wet ($\theta_a = 65.1^\circ$ and $\theta_r =$
499 60.4°) at typical storage conditions (323 K, 25 MPa), which turned strongly hydrophobic ($\theta_a =$
500 145.7° and $\theta_r = 136.2^\circ$) in the presence of 10⁻² mol/L lignoceric acid concentration (longer
501 alkyl chain) at similar conditions of 1-year ageing duration (such a change is attributed to
502 chemisorption of organic acids, thus reduced structural trapping potential). However, optimum
503 alumina nanofluid concentration (0.25 wt%) have shown a significant reduction in CO₂-
504 wettability and turned them weakly water-wet ($\theta_a = 73.5^\circ$ and $\theta_r = 68.9^\circ$), thus, increased
505 structural trapping potential.

506 In a nutshell, it is crucial to fully comprehend the CO₂-wettability of cap-rock storage
507 formations in the presence of organic acids and nanofluids at typical storage conditions, as
508 lower structural trapping potential may be predicted, which can affect the feasibility of long
509 term CO₂ geo-storage projects. We thus recommend that CO₂ geo-schemes and simulation
510 strategies should account for the benefits of nano-energy applications into CO₂ geo-storage
511 formations to further de-risking the project viability of CO₂ geo-sequestration.

512

513 **Conflicts of Interest**

514 There are no conflicts to declare.

515

516 **Acknowledgments**

517 The first author acknowledges the scholarship (Research Training Program Stipend – 2018)
518 provided by Australian Government for his higher studies, as well as Curtin University for
519 supervision and resources.

520

521 **Supporting Information**

522 Supporting information accompanies this article, where Energy Dispersive Spectroscopy
523 (EDS) results for each pristine mica, organic-aged mica and nano-aged mica substrates (Table
524 S1), are included.

525

526

527

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529

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Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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